論文内容の要旨

1. Introduction

Metal oxide nanomaterials play important roles in many fields, such as physics, chemistry, materials science, and biology, due to their unique properties originated from the nanosize, large surface area, and quantum effect, etc. The physical and chemical properties, for example chemical reactivity, surface area, porosity, and band gap energy of nanomaterials, can be tuned by controlling their sizes in nano-level. To this end, much efforts has been paid to the synthesis of size- and morphology-controlled nanomaterials by chemical methods, such as sol-gel, precipitation, microemulsion, and template synthesis. However, those chemical methods usually require multi-step reactions including long reaction time as well as calcination. Therefore, development of simple synthetic methods to yield size, composition, and morphology controlled nanomaterials is quite of importance.

Previously, Wang et al. reported a simple, rapid, one-pot, single-step, and template free synthetic method of spherical mesoporous assemblies of TiO₂ nanoparticles called micro/mesoporously architected roundly integrated metal oxides (MARIMOs) due to their similarity in shape to marimo (cladophora aegagropila) moss balls.¹⁾ where a large number of fine primary nanoparticles ranging in the size 5-10 nm assembled affording a porous spherical secondary structure with a large specific surface area. In solvothermal method, a precursor this solution of a metal alkoxide and a carboxylic acid in methanol is simply treated under high-



Figure 1. Synthesis of solid and hollow MARIMO assemblies by one-pot single-step method.

temperature and high-pressure conditions in a closed tubular reactor. TiO₂ solid and hollow MARIMO assemblies with narrow size distribution were easily synthesized by controlling the reaction parameters such as a sort of carboxylic acid, heating rate, and reaction temperature (Figure 1). Solid MARIMO families of SiO₂, ZnO, ZrO₂, and CeO₂ were also obtained according to the similar treatments of suitable precursors.

On the other hand, physical, chemical, and electrical properties of metal oxide materials can be improved by nano-level mixing of several metal oxides, where the stoichiometries, particle sizes, nano-structures, and morphologies are key factors controlling the properties of the resulting composite nanomaterials. In my study, I synthesized Al_2O_3 -TiO₂, ZnO-TiO₂ and CeO₂-ZrO₂ composite assemblies with different distributions of primary nanoparticles such as homogeneous, domain, and core-shell structures. Moreover, MgO ultra-fine nanocrystals were synthesized by choosing suitable magnesium salts, solvents, organic additives, and reaction temperature.

2. A simple solvothermal approach to Al₂O₃-TiO₂ and ZnO-TiO₂ mesoporous hollow composite assemblies.

Mesoporous hollow nanomaterials containing composite metal oxides such as Al_2O_3 -TiO₂ and ZnO-TiO₂ are quite of interest due to their intriguing morphology capable of material

accommodation and the advantageous synergistic effects between two metal oxides. Al₂O₃-TiO₂ composite materials are reported to exhibit high heat tolerance for the anatase-to-rutile phase transition, and ZnO-TiO₂ composite materials are used for band gap energy tuning. However, the synthesis of Al₂O₃-TiO₂ and ZnO-TiO₂ porous composite assemblies with hollow spherical morphologies permitting the effective control of their compositions has not been intensively studied. In this chapter, one-pot, single-step, and template-free solvothermal synthetic method to lead Al₂O₃-TiO₂ and ZnO-MARIMO hollow spherical TiO₂ assemblies consisting of Al₂O₃ and ZnO fine primary nano-particles mixed with TiO₂ domains, respectively, is discussed.

Al₂O₃-TiO₂ composite assemblies were synthesized by slowly heating (5.4 °C/min, up to 300 °C) of precursor solutions consisting of $Al(O^{i}Pr)_{3}$, Ti(O^{*i*}Pr)₄, and phthalic acid in methanol with different Al/Ti mixing ratios. The obtained Al₂O₃-TiO₂ hollow composite assemblies are referred to as Al/Ti-0.75, Al/Ti-0.50, Al/Ti-0.25, Al/Ti-0.20, Al/Ti-0.10, and Al/Ti-0.05, where the numbers denote the molar portion of $Al(O^{i}Pr)_{3}/(Al(O^{i}Pr)_{3} + Ti(O^{i}Pr)_{4})$ in the precursor solutions. The STEM/EDX mapping images of Al/Ti-0.20 indicate the even distribution of Al and Ti atoms throughout the hollow composite MARIMO assemblies (Figure 2a). A linear relationship between the atomic



Figure 2. STEM and EDX mapping images of (a) Al/Ti–0.20 and (b) Zn/Ti–0.20 hollow MARIMO composite assemblies.



Figure 3. XRD patterns (left) and TEM images (right) of (a) Al/Ti–0.50, (b) Al/Ti–0.25, and (c) Al/Ti–0.05 hollow composite assemblies after calcination at 1,000 °C for 1 h.

compositions of the Al₂O₃-TiO₂ hollow composite assemblies (determined by EDX) and the molar ratios of the precursor solutions was observed, indicating that the composition of the

 Al_2O_3 -TiO₂ hollow composite assemblies can be easily controlled by adjusting the Al/Ti mixing ratio in the precursor solutions. XRD patterns showed that the synthesized porous composite assemblies consist of anatase TiO₂ and amorphous Al₂O₃. In particular, it is notable that the BET specific surface area of Al/Ti–0.75 and Al/Ti–0.50 exceeded 600 m² g⁻¹.

The Al_2O_3 -TiO₂ hollow composite assemblies were calcinated at 1,000 °C for 1 h to evaluate their heat properties. The XRD patterns and TEM images of calcinated Al_2O_3 -TiO₂ composite assemblies are shown in Figure 6. Al/Ti-0.05 clearly



Figure 4. Plots of band gap energies of $ZnO-TiO_2$ hollow composite assemblies versus the initial molar fractions of Zn in precursor solutions.

showed the anatase-to-rutile phase transition and hollow spherical morphology was damaged (Figure 3c). The partial formation of the rutile phase was observed for Al/Ti–0.25 under similar conditions, while the anatase phase still remained (Figure 3b). However, in the case of Al/Ti–0.50, the anatase phase survived completely with no rutile phase formation. The hollow composite assemblies calcinated at 1,000 °C clearly indicate that the spherical hollow morphology of composite assemblies with Al = 25–50% remains intact even after the calcination at 1,000 °C for 1h (Figure 3a).

ZnO–TiO₂ hollow composite assemblies referred to as Zn/Ti–0.75, Zn/Ti–0.50, Zn/Ti–0.25, Zn/Ti–0.20, Zn/Ti–0.10, and Zn/Ti–0.05, where each number denotes the molar portion of $Zn(OAc)_2/(Zn(OAc)_2 + Ti(O^iPr)_4)$ in the precursor solutions, were also obtained by a similar solvothermal method. STEM/EDX mapping images of Zn/Ti–0.20 (Figure 2b) indicate the even distribution of Zn and Ti atoms throughout the MARIMO hollow composite assemblies. The band gap energies of ZnO–TiO₂ MARIMO composite assemblies were tunable by changing the ZnO/TiO₂ mixing ratio in the assemblies. The band gap energies of the as prepared ZnO–TiO₂ composites were around 3.3 eV. However, the calcination at 500 °C for 1 h caused relatively large shifts in band gap energies to lower values of 3.12-3.21 eV (Figure 4).

3. Synthesis of CeO₂–ZrO₂ mixed mesoporous spherical nanomaterials with homogeneous, domain, and core-shell morphologies.

Much attention has been paid to CeO_2 nanomaterials for their many applications such as oxygen storage, polishing materials, catalysts, and catalyst supports. However, decreasing the amount of the rare earth metal Ce without losing its performance has become a crucial aspect of current research because of the Ce resource limitation. The synthesis of CeO_2 –ZrO₂ composites is one approach to minimize Ce loading on nanomaterials. In this chapter, a simple and rapid synthetic approach to CeO_2 –ZrO₂ MARIMO composite assemblies with homogeneously dispersed, domain, and core-shell structures having mesoporous spherical morphologies is discussed.

CeO₂–ZrO₂ MARIMO composite assemblies were synthesized by a similar solvothermal approach treating precursor solutions containing Ce(NO₃)₃ and ZrO(NO₃)₂ with formic acid in methanol by means of rapid heating (500 °C/min, up to 300°C). These MARIMO composite assemblies referred to as Ce/ZrO₂–0.25, Ce/ZrO₂– 0.33, Ce/ZrO₂–0.50, Ce/ZrO₂–0.66, where the number designates the molar

portion $[Ce^{3+}]/([Ce^{3+}] + [Zr^{4+}])$ in the precursor solutions. The TEM and EDX mapping images (Figure 5a) of Ce/ZrO₂-0.50 sample indicated the formation of MARIMO assemblies in which Ce and Zr atoms were evenly distributed. In the XRD patterns of the Ce/ZrO₂ MARIMO assemblies, the continuous shifting of the diffraction peak positions and lattice parametrs from those of CeO_2 to ZrO_2 , depending on the Ce/Zr ratio in the precursor solutions (Figuress 6a-f), was observed. Judging from the TEM images and XRD patterns, the synthesis of homogenious Ce_xZr_{1-x}O₂ MARIMO assemblies were accomplished and the control of their compositions were sucessfully achieved by this simple method.

On the other hand, the sample CeO₂-ZrO₂-SH were obtained by slow heating (5.4 °C/min, up to 300 °C) of an equimolar mixture of Ce(NO₃)₃ and $ZrO(NO_3)_2$ with formic acid in methanol. The EDX mapping images showed that Ce, Zr, and O atoms were evenly distributed in CeO₂-ZrO₂-SH MARIMO composite assemblies (Figure 5b), while the XRD exhibited a mixed profile of cubic CeO_2 and tetragonal ZrO_2 phases (Figure 6g), indicating that CeO₂-ZrO₂-SH



Figure 5. STEM and EDX mapping images of (a) Ce/ZrO₂-0.50 and (b) CeO₂-ZrO₂-SH MARIMO composite assemblies.



Figure 6. XRD patterns of MARIMO NPs: (a) ZrO_2 , (b) Ce/ZrO₂-0.25, (c) Ce/ZrO₂-0.33, (d) Ce/ZrO₂-0.50, (e) Ce/ZrO₂-0.66, (f) CeO₂, and (g) CeO₂-ZrO₂-SH



Figure 7. EDX (a) mapping images and (b) line scans of $ZrO_2@CeO_2$ core-shell MARIMO composite assemblies.

MARIMO composite assemblies consist of a *domain* structure with cubic CeO_2 and tetragonal ZrO_2 .

A step-wise synthetic approach yielded $ZrO_2@CeO_2$ *core-shell* MARIMO assemblies with a ZrO_2 core and a CeO₂ shell (Figure 7) by the high-temperature and high-pressure treatment of ZrO_2 MARIMO assemblies with a solution of Ce(NO₃)₃ in 2-propanol.

4. Synthesis of Mg-nanomaterials

Mg-nanomaterials were successfully synthesized by our solvothermal method.

5. Conclusions

A simple synthetic method to afford Al_2O_3 -TiO₂ and ZnO-TiO₂ hollow composite assemblies was developed. Atomic compositions in Al_2O_3 -TiO₂ and ZnO-TiO₂ hollow composite assemblies were controllable through the mixing ratio of the corresponding metal sources in the precursor solutions. Al_2O_3 -TiO₂ hollow composite assemblies with high Al content (Al = 50%) did not show any phase transition from anatase-to-rutile even after calcination at 1,000 °C for 1 h. The band gap energy of TiO₂ in ZnO-TiO₂ assemblies was tuned by changing their ZnO content. A simple synthetic method to afford CeO₂-ZrO₂ porous spherical composite assemblies was also developed. Homogeneous, domain, and core-shell structures of CeO₂-ZrO₂ porous spherical composite assemblies were easily obtained by tuning the solvothermal conditions of solvents, organic additives, and heating rate. Atomic compositions in homogenous CeO₂-ZrO₂ porous spherical composite assemblies were also controllable through the Ce/Zr mixing ratio in precursor solutions. Moreover, Mg-nanomaterials were successfully synthesized.

6. References

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List of works

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- 2. E. K. C. Pradeep, M. Ohtani, K. Kobiro, A simple synthetic approach to Al₂O₃-TiO₂ and ZnO-TiO₂ mesoporous hollow composite assemblies consisting of homogeneously mixed primary particles at the nano level, *Eur. J. Inorg. Chem.*, **2015**, 5621–5627.

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- 5. E. K. C. Pradeep, M. Ohtani, K. Kobiro, One-pot Single-step Synthetic Approach to Morphology Controlled Binary Metal Oxide Nanospheres, *Kochi University-Kochi University of Technology Joint Seminar of Chemistry*, Kochi, Japan, June 2015.
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Awards

1. Journal of Supercritical Fluids "Editor-in-chief's Featured Article" award,

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